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UTILITY PATENT APPLICATION TRANSMITTAL <small>(Only for new nonprovisional applications under 37 CFR 1.53(b))</small>	Attorney Docket No. 5576-131
	First Inventor or Application Identifier: Maeda et al.
	Title of Invention: HYDROGEN ABSORBING ALLOY AND NICKEL-METAL HYDRIDE RECHARGEABLE BATTERY
	Express Mail Label No. EL618192015US

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Transmitted herewith for filing in the United States Patent Office is a patent application for:

Inventors: Takao Maeda; Satoshi Shima; Naofumi Shinya

1. ☒ The Filing Fee has been calculated as shown below:

	No. Filed	No. Extra	Small Entity		Large Entity	
			Rate	Fee 0	Rate	Fee 1
BASIC FEE				\$0		\$690
TOTAL CLAIMS:	12 - 20 =	0		X 9 = \$0		x 18 = \$0
INDEP CLAIMS:	4 - 3 =	1		X 39 = \$0		x 78 = \$78
<input type="checkbox"/> MULTIPLE DEPENDENT CLAIMS PRESENTED				+130 = \$		+260 = \$
*If the difference in Column 1 is less than zero, enter "0" in Column 2.				TOTAL \$		TOTAL \$ 768

The Commissioner is hereby authorized to credit overpayments or charge the following fees to Deposit Account No. 16-0605.

- a. ☒ Fees required under 37 CFR 1.16 (National filing fees).
b. ☒ Fees required under 37 CFR 1.17 (National application processing fees).
☒ A check in the amount of \$ 768.00 for the filing fee is enclosed.
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2. ☒ Specification; Total Pages 27
3. ☒ 1 Sheets of Formal Drawing(s) (35 USC 113)

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4. ☒ Declaration and Power of Attorney; [Total Pages 3]
- a. ☒ Newly executed (original or copy)
- b. ☐ Copy from a prior application (37 CFR 1.63(d))
(for continuation/divisional with Box 16 completed)
- i. ☐ DELETION OF INVENTOR(S) Signed statement
attached deleting inventor(s) named in the prior
application, see 37 CFR 1.63(d)(2) & 1.33(b).
5. ☐ Microfiche Computer Program (Appendix)
6. ☐ Nucleotide and/or Amino Acid Sequence Submission (if applicable, all necessary)
- a. ☐ Computer Readable Copy
- b. ☐ Paper Copy (identical to computer copy)
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ACCOMPANYING APPLICATION PARTS

7. ☒ Assignment Papers (cover sheet & document(s) (including a check for
the \$40.00 fee)
8. ☐ 37 CFR 3.73(b) Statement (when there is an assignee); ☐ Power of Attorney
9. ☐ English Translation Document (if applicable)
10. ☐ Information Disclosure Statement (IDS)/PTO-1449; ___ Copies of IDS Citations
11. ☒ Preliminary Amendment
12. ☒ Return Receipt Postcard (MPEP 503) (Should be specifically itemized)
13. ☐ Small Entity Statement(s)
☐ Statement filed in prior application; status still proper and desired.
14. ☐ Certified Copy of Priority Document(s) (if foreign priority is claimed)
Foreign Priority is
15. ☐ Other: ___

16. **If a CONTINUING APPLICATION, check appropriate box and supply the requisite
information below and in a preliminary amendment:**

☐ Continuation ☐ Divisional ☐ Continuation in Part (CIP)
of prior Application No: ___; Filed ___

Prior Application Information: Examiner

Group/Art Unit:

For CONTINUATION or DIVISIONAL APPS only. The entire disclosure of the prior application, from which an oath or declaration is supplied under Box 4b, is considered a part of the disclosure of the accompanying continuation or divisional application and is hereby incorporated by reference. The incorporation can only be relied upon when a portion has been inadvertently omitted from the submitted application parts.

17. **CORRESPONDENCE ADDRESS**Customer Number or Bar Code Label **000826**

Attention Of: Melissa B. Pendleton

Signature:

Melissa B. Cudde

Attorney/Agent of Record: Melissa B. Pendleton

Attorney/Agent Registration No. 35,459

Tel Charlotte Office (704) 331-6000

Fax Charlotte Office (704) 334-2014

ALSTON & BIRD LLP

P.O. Drawer 34009

Charlotte NC 28234-4009

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Grace R. Rön

Grace R. Rippey

IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

In re: Maeda et al.
Filed: Concurrently Herewith
For: HYDROGEN ABSORBING ALLOY AND
NICKEL-METAL HYDRIDE RECHARGEABLE BATTERY

August 3, 2000

Assistant Commissioner for Patents
Washington, DC 20231

PRELIMINARY AMENDMENT

Sir:

Please amend the above-identified application as follows:

In The Claims:

1. (Amended) A hydrogen absorbing alloy having a CaCu_5 type crystal structure in its principal phase, comprising La in [the range] an amount of 24 to 33% by weight in the alloy, and Mg or Ca in [the range] an amount of 0.1 to 1.0% by weight in the alloy.

4. (Amended) A hydrogen absorbing alloy according to claim 1, further comprising [wherein the] Co [content is] in an amount of 6 to 9% by weight, and having an [the] atomic ratio B/A [is] of 5.0 to 5.25, wherein A represents a rare earth element including La, and B represents a rare earth element, transition metal or Al.

In claim 5, line 2, please insert -- elements -- before "selected".

In claim 7, lines 1 and 2, please delete "any one of claims 1 to 4" and insert therefor -- claim 1 --.

Cancel claims 8, 9, 10, and 11.

12. A hydrogen absorbing alloy according to claim 1, wherein said alloy is

R is a rare earth element other than La;

the ratio of $(x+y+z)/(u+v)$ is 4 to 7.

forming a melt of elemental components selected to form a hydrogen absorbing alloy; and

14. A method for manufacturing a hydrogen absorbing alloy according to claim 13,

15. A method for manufacturing a hydrogen absorbing alloy according to claim 14,

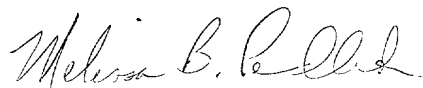
16. A nickel-metal hydride rechargeable battery comprising an electrode formed of a

hydrogen absorbing alloy having a CaCu_5 type crystal structure in its principal phase, said alloy comprising La in an amount of 24 to 33% by weight in the alloy, and Mg or Ca in an amount of 0.1 to 1.0% by weight in the alloy.

Remarks

The claims have been amended to place them in better form for examination in accordance with standard U.S. practice, and additional claims have been presented for examination. Please enter this amendment prior to calculation of the filing fee.

Respectfully submitted,

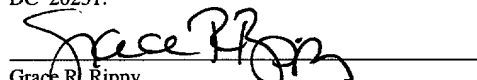


Melissa B. Pendleton
Registration No. 35,459

ALSTON & BIRD LLP
P.O. Drawer 34009
Charlotte, NC 28234
Tel Charlotte Office (704) 331-6000
Fax Charlotte Office (704) 334-2014

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Grace R. Rippey

TITLE OF THE INVENTION

Hydrogen Absorbing Alloy and Nickel-Metal hydride
Rechargeable Battery

BACKGROUND OF THE INVENTION

5 1. Field of the invention

This invention relates to hydrogen absorbing alloys, and more particularly to a hydrogen absorbing alloy which can be used to form negative electrodes for use in nickel-metal hydride rechargeable (secondary) batteries.

10 2. Description of the related art

Conventionally, misch metal (hereinafter referred to as "Mm") comprising a mixture of rare earth elements such as La, Ce, Pr, Nd and Sm, and nickel-base alloys formed by replacing a part of Ni with various elements are widely used as
15 hydrogen absorbing alloys for forming negative electrodes for use in nickel-metal hydride rechargeable batteries.

It is known that, among others, cobalt-containing alloys are capable of absorbing a relatively large amount of hydrogen, are less liable to particle size reduction in their
20 hydrogen-loaded state, have excellent corrosion resistance in alkalis, and are effective in prolonging the lives of nickel-metal hydride rechargeable batteries when they are used for the negative electrodes thereof.

On the other hand, it is also known that lower cobalt
25 contents are more desirable for an improvement of a high rate

discharge property. The reason for this is believed to be that a decrease in cobalt content promotes particle size reduction and hence causes an increase in surface area per unit weight.

5 SUMMARY OF THE INVENTION

10 In order to solve these problems of the prior art, the present invention provides a hydrogen absorbing alloy which can improve a high rate discharge property while suppressing particle size reduction, exhibits cycle life characteristics equal to or higher than those of conventional alloys even when its cobalt content is decreased, and has a high capacity.

15 The present invention is based on the discovery that, when a hydrogen absorbing alloy has a relatively high La content and contains an alkaline earth metal (i.e., Mg or Ca) in a relatively small amount above impurity levels, the alloy can improve a high rate discharge property in spite of suppressed particle size reduction while maintaining its high capacity, and can suppress particle size reduction even when
20 its cobalt content is decreased to less than the conventionally known level.

Specifically, the present invention relates to a hydrogen absorbing alloy having a CaCu₂ type crystal structure in its principal phase, wherein the La content in
25 the alloy is in the range of 24 to 33% by weight and the Mg

or Ca content in the alloy is in the range of 0.1 to 1.0% by weight.

In a preferred embodiment, the present invention also relates to the aforesaid alloy wherein the cobalt content in the alloy is not greater than 9% by weight.

When the hydrogen absorbing alloy of the present invention is used for the negative electrode of an alkaline rechargeable battery, it can increase the capacity of the battery, can improve the high rate discharge property thereof, and can suppress particle size reduction even at low cobalt contents to cause a reduction in battery cost.

BRIEF DESCRIPTION OF THE DRAWING

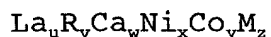
FIG.1 is a x-ray diffraction pattern for the hydrogen absorbing alloy of Example 1.

DETAILED DESCRIPTION OF THE PREFERRED EMBODIMENTS

In the AB₅ type hydrogen absorbing alloy of the present invention, 0.1 to 1.0% by weight of Mg or Ca is contained in order to improve the high rate discharge property while suppressing particle size reduction. Moreover, the La content in the alloy is set at 24 to 33% by weight in order to increase the amount of hydrogen absorbed and control the equilibrium pressure of hydrogen. Thus, as contrasted with conventional alloys, the hydrogen absorbing alloy of the present invention has a high capacity, can improve the high rate discharge property while suppressing particle size

reduction, and can enhance resistance to particle size reduction even at low cobalt contents.

Such hydrogen absorbing alloys may, for example, be expressed in terms of the following chemical formulas:



wherein R is a rare earth element other than La, M is at least one element chosen from the group consisting of Mn, Al, Si, Sn, Fe, Cu, Ti, Zr, and V or the like, the content of La is preferably 24 to 33 % by weight, R 15 % by weight, Mg or Ca 0.1 to 1.0 % by weight, Ni 50 to 60 % by weight, Co 9 % by weight or less, and M 3 to 10 % by weight. Here, the compositional ratios of elements are expressed in terms of atomic ratios (u, v, w, x, y and z). These atomic ratios can be obtained by dividing the percentage-by-weight for each element by the respective atomic weight and then by normalizing the resulting figures using the sum of constitutional ratios of La and R, which are classified as "A" elements. Thus, u plus v equals 1 by definition.

20 Because R is a rare earth element which is other than La, and M is at least one element chosen from the group of Mn, Al, Si, Sn, Fe, Cu, Ti, Zr, V or the like, the weighted averages of atomic weights are used for R and M. Excluding Mg and Ca, which are added in minor amounts, as well as unavoidable

25 impurities, the ratio of elements belonging to "B" to those

belonging to "A" is calculated as a B/A ratio according to the following equation: $B/A \text{ ratio} = (x+y+z)/(u+v)$.

Furthermore, in the AB_5 type hydrogen absorbing alloy of the present invention, the remainder of the moiety A

5 comprises one or more rare earth elements other than La, and the remainder of the moiety B comprises one or more transition metals such as Ni, Co and Mn and/or Al or the like. The atomic ratio of B to A, B/A, is preferably 4 to 7, more preferably 5 to 7, further more preferably 5 to 6.

10 The AB_5 type hydrogen absorbing alloy used in the present invention is preferably a hydrogen absorbing alloy having a $CaCu_5$ type crystal structure in its principal phase. As used herein, the expression "hydrogen absorbing alloy having a $CaCu_5$ type crystal structure in its principal phase"

15 refers to a hydrogen absorbing alloy in which, although segregation phases are partly recognized by metallographic observation of a section, the diffraction pattern recorded by XRD exhibits a $CaCu_5$ type alloy phase.

The hydrogen absorbing alloy of the present invention is

20 characterized in that its Mg or Ca content is in the range of 0.1 to 1.0% by weight. If its Mg or Ca content is less than 0.1% by weight, the effect of suppressing particle size reduction will be insufficient. If its Mg or Ca content is greater than 1.0% by weight, the amount of hydrogen absorbed

25 will be decreased to an undue extent.

When the Co content is lowered, the equilibrium pressure of hydrogen at the time of absorption or desorption of hydrogen is elevated. Accordingly, the La content is set at 24 to 33% by weight in order to maintain the equilibrium pressure of hydrogen at the same level as those of conventional alloys and to maintain or improve the high capacity. In the present invention, it is especially preferable to add Mg.

Moreover, the present invention involving the addition of a relatively small amount of Mg or Ca as described above makes it possible to achieve a long life at cobalt contents of not greater than 9% by weight, preferably 7% by weight or less, and more preferably 6% by weight or less, as contrasted with the prior art in which it has been unachievable.

The addition of a small amount of one or more selected from the group consisting of Ti, Zr and V to the Mg- or Ca-containing hydrogen absorption alloy can enhance the initial characteristics or cycle life characteristics. The amount of the addition is as small as 0.5% by weight or less based the Mg- or Ca-containing hydrogen absorbing alloy.

Moreover, the Mg-containing hydrogen absorbing alloy has a CaCu₂ type crystal structure in its principal phase, wherein the length for a-axis (a-axis = b-axis) is in the range of 4.990 to 5.050Å, the length for c-axis is in the range of 4.030 to 4.070, regarding the lattice constants

thereof. Comparing the lattice constants in these range between Mg-free and Mg-containing hydrogen absorbing alloys, the addition of Mg tends to increase the lattice constants. It has been particularly found that increase for c-axis is
5 larger than that for a-axis so that the ratio of length of c-axis to length of a-axis, c/a , becomes larger.

It has been found that increase in the ratio c/a results in less liability to particle size reduction so as to produce a battery with longer cycle life. The reason for this is
10 believed to be that the larger face distance between the face perpendicular to c-axis, which are the faces for closest packing of crystal, suppress the extension of the lattice. Consequently, the stress is restrained and the developing distances for cracks become smaller. Thus, the less
15 liability to the particle size reduction for the hydrogen absorbing alloy comprising 0.1 to 1.0% by weight of Mg is thought to be derived from c-axis having longer extension than a-axis.

Further, it has been found that, the hydrogen absorbing
20 alloy, having 24 to 33% by weight of La, 6 to 9% by weight of Co and the atomic ratio B/A of 5.0 to 5.25, and which Mg is added to in an amount of 0.1 to 1.0% by weight, can result in a battery with higher capacity such as 340mAh/g or more, keeping cycle life unchanged. In this case, the ratio B/A
25 means the sum of atomic ratios of, for example, Ni, Co, Mn

and Al, excluding the elements in very small amounts such as Mg and Ca, by designating the sum of atomic ratios of rare-earth metals such as La, Ce, Pr and Nd for one.

The hydrogen absorbing alloy of the present invention
5 can be manufactured by a dissolution method such as arc dissolution and high frequency dissolution, casting in a mold, table-casting, a rapid roll quenching method, gas atomization, disk-atomization or a spin-cup method, or a combination thereof.

10 The hydrogen absorbing alloy of the present invention may be prepared in the following manner.

Predetermined amounts of various elements may be weighed out and melted in a high-frequency furnace having an atmosphere of an inert gas (at 200 to 1,500 Torr) such as Ar
15 gas. In the case of an element (e.g., Mg or Ca) having a high vapor pressure, it may be added directly by itself, or in the form of an alloy formed of such an element and one or more other elements constituting the alloy. In the melting method, it may be preferable that Mg or Ca is not added until
20 metals with high melting point such as Ni or Co are melted in order to prevent added components from evaporating or assure the safe operation. The resulting melt may be cast in a mold made of iron at a temperature of 1,300 to 1,600°C to form an ingot. Or the other methods mentioned above may be also
25 used. In case of special need, the ingot may be heat-treated

at a temperature of 800 to 1,200°C for 5 to 20 hours in an inert atmosphere (at 600 to 1,500 Torr), for example, of Ar gas.

Using a jaw crusher, a roll mill, a hammer mill, a pin
5 mill, a ball mill, a jet mill, a roller mill and the like,
the hydrogen absorbing alloy prepared in the above-described
manner may be ground to an average particle diameter of 4 to
70 μm in an inert atmosphere, for example, of Ar. Moreover,
reduction in particle size by hydrogen absorption and
10 desorption, so-called hydrogenation method, may be used.
Thus, there can be obtained a hydrogen absorbing alloy in
accordance with the present invention.

The hydrogen absorbing alloy powder thus obtained may be
formed into electrodes according to any well-known method.
15 This can be accomplished, for example, by mixing the alloy
powder with a binder selected from polyvinyl alcohol,
cellulose derivatives (e.g., methylcellulose), PTFE,
polyethylene oxide, high polymer latices and the like,
kneading this mixture into a paste, and applying this paste
20 to an electrically conducting three-dimensional support
(e.g., foamed nickel or fibrous nickel) or an electrically
conducting two-dimensional support (e.g., punching metal).
The amount of binder used may be in the range of 0.1 to 20%
by weight per 100% by weight of the alloy.

Moreover, if necessary, an electrically conducting filler such as carbon-graphite powder, Ni powder or Cu powder may be added in an amount of 0.1 to 10% by weight based on the alloy.

5 Alkaline batteries using the hydrogen absorbing alloy of the present invention for the negative electrodes thereof have a long cycle life and exhibit an excellent high rate discharge property and low-temperature discharge characteristics, even when the alloy has a low cobalt
10 content.

The present invention is further illustrated by the following examples. However, these examples are not to be construed to limit the scope of the invention.

Example 1 and Comparative Example 1

15 Mm or rare earth elements such as La, Ce, Pr and Nd, metallic elements such as Ni, Co, Mn and Al, and Mg were weighed out so as to give each of the compositions shown in Table 1. Mg was used in the form of a MgNi₂ (m.p. 1100°C) alloy. These materials were melted in a high-frequency
20 melting furnace, and the resulting melt was cast in a mold made of iron to form an ingot. As for the Mg-free alloy, the ingot is formed without using the Mg-Ni alloy.

This ingot was heat-treated at 1,050°C for 6 hours in an atmosphere of Ar. Thereafter, using a grinder, this ingot
25 was ground to an average particle diameter of 33 μm so as to

obtain a hydrogen absorbing alloy powder. Analysis of this allow powder by XRD revealed that it had a CaCu_2 type crystal structure (FIG.1).

10 g of this powder was mixed with 2.5 g of a 3 wt% aqueous solution of polyvinyl alcohol (with an average degree of polymerization of 2,000 and a degree of saponification of 98 mole %) to prepare a paste. This paste was filled into a porous metallic body of foamed nickel in an amount of 30% by volume, dried, and then pressed into a plate having a thickness of 0.5-1.0 mm. Finally, a negative electrode was made by attaching a lead wire thereto.

A positive electrode, which comprised a sintered electrode, was bonded to the aforesaid negative electrode with a polypropylene separator interposed therebetween. This assembly was immersed in a 6N KOH electrolyte to construct a battery.

Each of the batteries so constructed was tested in the following manner. First of all, at a temperature of 20°C , the battery was charged to 120% at 0.3C (90mA/g) based on the capacity of the negative electrode, rested for 30 minutes, and then discharged at 0.2C (60mA/g) until the battery voltage reached 0.6 V. When this cycle was repeated twenty times, the greatest discharge capacity was regarded as the "capacity" of the alloy. Subsequently, this battery was charged to 120% at 0.3C, and discharged at 2.0C (600mA/g).

The capacity measured in this manner was regarded as the
 "high rate discharge capacity". Thereafter, in order to
 observe the degree of particle size reduction, the negative
 electrode was disassembled, placed in water, and exposed to
 5 ultrasonic waves from an ultrasonic horn so as to separate
 the alloy powder from the current collector. The particle
 size distribution after repeated charging and discharging was
 measured by means of a Microtrack analyzer to determine the
 average particle diameter D_{50} (μm). The results thus
 10 obtained are shown in Table 1. It is to be understood that,
 when the frequency of occurrence of each particle diameter in
 the measured particle size distribution is cumulatively added
 from the smaller to the larger side, the particle diameter
 corresponding to 50% of the entire distribution is defined as
 15 D_{50} .

Table 1

	Alloy composition (wt%)									Capacity (mAh/g)	High rate discharge capacity (mAh/g)	Average particle diameter after size reduction (μm)
	La	Ce	Pr	Nd	Mg	Ni	Co	Mn	Al			
Example 1	25.04	3.16	1.90	1.30	0.27	56.61	5.31	4.58	1.82	305	220	25.31
Comparative Example 1	25.45	3.21	1.94	1.32	0.00	53.76	8.64	3.77	1.92	302	162	23.14

As shown in Table 1, Mg-containing alloy has a better high rate discharge property and less liability to particle size reduction.

Examples 2-5 and Comparative Example 2

5 The compositions shown in Table 2 were employed for the formation of alloys in the same manner as Example 1 and capacities were measured in the same matter as Example 1 to examine the relationship between the La content and the capacity when magnesium is contained in the alloys. The
10 results thus obtained are shown in Table 2. It can be seen from Table 2 that, in order to obtain an alloy having a high capacity, the La content in the alloy must be not less than 24% by weight.

Table 2

	Alloy composition (wt%)									Capacity (mAh/g)
	La	Ce	Pr	Nd	Mg	Ni	Co	Mn	Al	
Example 2	25.56	3.87	1.30	1.33	0.17	58.86	2.71	3.79	2.42	306
Example 3	25.06	3.79	1.27	1.30	0.16	58.92	2.66	4.46	2.37	297
Example 4	24.86	3.76	1.26	1.29	0.27	59.22	2.64	3.69	3.02	293
Example 5	24.69	3.74	1.25	1.28	0.27	58.81	2.62	5.00	2.34	289
Comparative Example 2	23.80	6.25	1.32	1.35	0.29	57.44	2.77	4.38	2.41	275

Examples 6-8 and Comparative Example 3

Employing the compositions shown in Table 3, alloy powders were prepared in the same manner as in Example 1. Then, electrode tests were carried out in the same manner as in Example 1 to determine the respective capacities. The results thus obtained are shown in Table 3. It can be seen from Table 3 that Mg contents of greater than 1.0% by weight cause an undue reduction in capacity.

Table 3

	Alloy composition (wt%)									Capacity (mAh/g)
	La	Ce	Pr	Nd	Mg	Ni	Co	Mn	Al	
Example 6	26.59	3.87	1.30	1.33	0.17	58.94	2.71	3.80	2.30	306
Example 7	25.53	3.86	1.29	1.33	0.28	58.80	2.71	3.79	2.42	301
Example 8	24.97	3.78	1.27	1.30	0.55	58.69	2.65	4.44	2.36	286
Comparative Example 3	24.99	3.15	1.27	1.30	1.09	58.74	2.65	4.45	2.37	270

Examples 9-12 and Comparative Examples 4-7

Employing the alloy compositions shown in Table 4, electrode tests were carried out in the same manner as in Example 1. Thereafter, each negative electrode was disassembled, placed in water, and exposed to ultrasonic waves from an ultrasonic horn so as to separate the alloy powder from the current collector. The particle size

distribution after repeated charging and discharging was measured by means of a Microtrack analyzer to determine the average particle diameter D_{50} (μm). On the basis of the average particle diameter of an alloy containing no Mg, the effect of Mg addition, i.e. the improvement of particle size reduction, was calculated as R1 (%) according to the following equation.

$$R1(\%) = \left\{ \frac{(D_{50} (\mu\text{m}) \text{ of Mg-containing alloy})}{(D_{50} (\mu\text{m}) \text{ of Mg-free alloy})} \right\} \times 100 (\%)$$

Since the degree of particle size reduction varies greatly with the Co content, the improvement of particle size reduction is independently shown with respect to each Co content. The D_{50} is defined in such a way that, when the particle size distribution of the hydrogen absorbing alloy is measured and the frequencies of detection of various particle diameters are cumulatively added from smaller-diameter to larger-diameter particles, the particle diameter corresponding 50% of all particles is represented by D_{50} .

Table 4

	Alloy composition (wt%)									Improvement of particle size reduction R1 (%)
	La	Ce	Pr	Nd	Mg	Ni	Co	Mn	Al	
Example 9	25.53	3.86	1.29	1.33	0.28	58.80	2.71	3.79	2.42	129.2
Example 10	24.97	3.78	1.27	1.30	0.55	58.69	2.65	4.44	2.36	134.7
Comparative Example 4	25.60	3.87	1.30	1.33	0.00	58.96	2.72	3.80	2.42	100.0
Comparative Example 5	25.56	3.87	1.30	1.33	0.08	58.86	2.71	3.79	2.42	104.0
Example 11	25.51	3.86	1.29	1.32	0.28	56.19	5.41	3.78	2.35	121.7
Comparative Example 6	25.58	3.87	1.30	1.33	0.00	56.34	5.43	3.79	2.36	100.0
Example 12	25.38	3.84	1.29	1.32	0.28	53.62	8.61	3.76	1.91	110.5
Comparative Example 7	25.45	3.85	1.29	1.32	0.00	53.77	8.64	3.77	1.92	100.0
Example 12	25.37	3.84	1.29	1.32	0.28	53.20	9.42	3.39	1.91	103.0
Comparative Example 8	25.44	3.85	1.29	1.32	0.00	53.09	9.45	3.65	1.92	100.0

It can be seen from Table 4 that the addition of Mg suppresses particle size reduction at the same Co content, and this effect becomes more pronounced as the Co content is decreased. It can also be seen that, at a low Mg content, for example, of less than 0.1% by weight, the improvement of particle size reduction is as low as 5% or less. Moreover, it can also be seen that, at a high Co content, for example, of greater than 9% by weight, the effect of Mg addition is lessened. In commercially available nickel-metal hydride

rechargeable batteries having a high capacity, the Co content is usually not less than 9%. However, it can be seen that the present invention exhibits a significant effect at Co contents of not greater than 7%.

5 Examples 14-17 and Comparative Examples 8-11

Employing the alloy compositions shown in Table 5, the alloy powders were prepared in the same manner as in Example 1 except the following: metallic Mg (m.p. 650°C) was used instead of the Mg-Ni alloy, and the mixture of Ni, Co, Mn, Al and some of rare-earth elements were melted in advance, and then after confirming the melting, the other of rare-earth elements and metallic Mg were added. As for the Mg-free alloys, the melting was carried out without addition of the metallic Mg.

15 The capacity in Table 5 was measured as follows. After dry-mixing hydrogen absorbing alloy 0.5 and Ni power 1.5 in the weight ratio, the mixture was molded in a mold with a diameter of 20mm to produce an electrode. The battery was charged to 125% at 0.5C (150mA/g), rested for 10 minutes, and then discharged at 0.5C (150mA/g) until the voltage difference based on mercury reference electrode (Hg/HgO) reached 0.6V. After this cycle was repeated ten times, the capacity was measured (as pellet capacity).

Moreover, Cycle life was measured as follows. Using the above-mentioned sample battery having the paste electrode, at

a temperature of 20°C, the battery was charged to 120% at 0.3C (90mA/g) based on the capacity of the negative electrode, rested for 30 minutes, and then discharged at 0.2C (60mg/g) until the battery voltage based on the positive electrode reached 0.8V. This cycle for charge and discharge was repeated two hundred times, and the maintenance of discharge capacity (cycle life) was calculated using the next equation.

Maintenance(%)

$$10 = \{(\text{discharge capacity after 200 cycles})/(\text{discharge capacity after 20 cycles})\} \times 100$$

Further, using the above-mentioned sample battery having the paste electrode, at a temperature of 20°C, the battery was charged at 0.3C (90mA/g) based on the capacity of the negative electrode, rested for 30 minutes, and then discharged at 0.2C (60mA/g) until the battery voltage reached 0.8V. After this cycle was repeated twenty times, in order to observe the degree of particle size reduction, the battery was disassembled and the alloy powder for the negative electrode was exposed to ultrasonic waves from an ultrasonic horn so as to separate the alloy powder from the current collector. The particle size distribution after repeated charging and discharging was measured by means of a Microtrack analyzer to determine the average particle

diameter D_{50} (μm). The improvement of particle size reduction R1 was calculated.

The diffraction patterns for the alloys shown in Table 5 were measured using a X-ray diffraction method for powder.

- 5 The lattice constants were calculated based on the measurement data using a method of least squares.

Table 5

	Alloy composition (wt%)								Capacity (mAh/g)	Cycle life (%)	Improvement of particle size reduction R1 (%)	Length of a-axis (Å)	Length of c-axis (Å)	Elongation of a-axis (Å)	Elongation of c-axis (Å)	
	La	Ce	Pr	Nd	Mg	Ni	Co	Mn								Al
Example 14	25.07	3.79	1.27	1.30	0.27	57.88	3.99	4.46	1.95	330	90	117	5.020	4.063	0.002	0.007
Comparative Example 8	25.14	3.80	1.28	1.31	0.00	58.04	4.00	4.48	1.95	320	82	100	5.018	4.056	Reference	Reference
Example 15	25.04	3.79	1.27	1.30	0.27	56.36	5.31	4.83	1.82	335	93	114	5.024	4.061	0.001	0.006
Comparative Example 9	25.11	3.80	1.27	1.30	0.00	56.51	5.33	4.84	1.83	330	85	100	5.023	4.055	Reference	Reference
Example 16	28.07	3.86	0.00	0.00	0.28	55.81	6.77	3.66	1.55	350	83	115	5.033	4.048	0.000	0.008
Comparative Example 10	28.15	3.87	0.00	0.00	0.00	55.97	6.79	3.67	1.55	330	70	100	5.033	4.040	Reference	Reference
Example 17	28.07	3.86	0.00	0.00	0.28	54.59	8.12	3.53	1.55	348	90	107	5.034	4.046	0.001	0.006
Comparative Example 11	28.15	3.87	0.00	0.00	0.00	54.74	8.14	3.54	1.55	325	80	100	5.033	4.040	Reference	Reference

As shown in Table 5, focusing on the effects of the addition of Mg, the addition of Mg increases capacity, cycle life, and improvement of particle size reduction. Comparison of the lattice constants shows that the addition of Mg tends to increase the c-axis more remarkably than a-axis. This is thought to be one of the reasons why the higher capacity and increased cycle life are attained. The results for Examples 16 and 17 show the specific increase of discharge capacity, although increase of cycle life therefor is fair.

10 Examples 18 to 32, Comparative Examples 12 to 19

Employing the alloy compositions shown in Table 6, the alloy powders were prepared using MgNi_2 (m.p. 1100°C) in the same manner as in Example 1 except the following: the mixture of Ni, Co, Mn, Al and some of rare-earth elements were melted at first. Then, after the confirmation of the melting, the other of rare-earth elements and the Mg-Ni alloy were added for melting. As for the Mg-free alloys, the melting was carried out without addition of the metallic Mg.

Pellet capacity and maintenance of discharge capacity (cycle life) were obtained as the same manner as described above. After the average particle diameter D_{50} was obtained as the same manner as described above, the improvement of particle size reduction was calculated as R2 (%) according the following equation. The R2 shows the inhibition effect against particle size reduction for the alloys other than the

alloy for Example 16 on the basis of the average particle diameter of the alloy of Comparative Example 16.

$$R2(\%) = \{(D_{50} (\mu\text{m}) \text{ of Mg-containing alloy}) / (D_{50} (\mu\text{m}) \text{ of the alloy for Example 16 alloy})\} \times 100 (\%)$$

5

Table 6

	Alloy composition (wt%)									Ratio B/A	Capacity (mAh/g)	Cycle life (%)	Improvement of particle size reduction R2 (%)
	La	Ce	Pr	Nd	Mg	Ni	Co	Mn	Al				
Example 18	28.74	3.22	0.00	0.00	0.28	53.30	8.81	3.79	1.86	5.20	340	93	137
Example 19	28.74	3.22	0.00	0.00	0.28	53.98	8.13	3.79	1.86	5.20	350	90	130
Example 20	28.74	3.22	0.00	0.00	0.28	54.66	7.45	3.79	1.86	5.20	350	88	130
Comparative Example 12	28.82	3.23	0.00	0.00	0.00	52.77	9.51	3.80	1.87	5.20	315	90	130
Comparative Example 13	28.82	3.23	0.00	0.00	0.00	53.45	8.83	3.80	1.87	5.20	320	83	130
Comparative Example 14	28.82	3.23	0.00	0.00	0.00	54.13	8.15	3.80	1.87	5.20	323	77	125
Comparative Example 15	28.82	3.23	0.00	0.00	0.00	54.81	7.47	3.80	1.87	5.20	328	73	115
Comparative	28.82	3.23	0.00	0.00	0.00	55.49	6.79	3.80	1.87	5.20	330	89	100
Example 16	28.74	3.22	0.00	0.00	0.28	52.63	9.48	3.79	1.86	5.20	325	97	140
Example 21	28.74	3.22	0.00	0.00	0.28	55.33	6.77	3.79	1.86	5.20	350	83	120
Example 22	30.34	0.97	0.32	0.33	0.28	53.98	8.13	3.79	1.86	5.20	350	88	130
Example 23	28.74	2.58	0.32	0.33	0.28	53.97	8.13	3.79	1.86	5.20	350	90	130
Example 24	27.14	3.86	0.65	0.33	0.28	53.97	8.13	3.79	1.86	5.20	345	92	130
Example 25	23.93	6.44	0.97	0.66	0.28	53.94	8.12	3.79	1.86	5.20	340	92	130
Comparative	32.03	0.00	0.00	0.00	0.00	54.15	8.15	3.80	1.87	5.20	350	75	110
Example 17	21.13	7.31	1.23	6.27	0.00	51.03	7.69	3.58	1.76	5.20	330	92	130
Comparative Example 18	20.79	9.04	1.30	1.00	0.00	54.07	8.14	3.80	1.86	5.20	320	93	130
Comparative Example 19	27.89	3.84	0.00	0.00	0.28	56.15	6.70	3.62	1.53	5.25	345	85	121
Example 27	28.07	3.86	0.00	0.00	0.28	55.86	6.74	3.64	1.54	5.20	350	85	118
Example 28	28.26	3.89	0.00	0.00	0.28	55.56	6.79	3.67	1.55	5.15	350	83	114
Example 29	28.46	3.91	0.00	0.00	0.28	55.26	6.83	3.69	1.56	5.10	355	80	110
Example 30	28.65	3.94	0.00	0.00	0.28	54.95	6.88	3.72	1.57	5.05	355	75	107
Example 31	28.85	3.97	0.00	0.00	0.29	54.64	6.93	3.75	1.59	5.00	360	70	105

As shown in Table 6, the alloys keeping La content of 24 to 33% by weight and Co content of 6 to 9% by weight in addition of 0.1 to 1.0% by weight of Mg with the B/A atomic ratio of 5.0 to 5.25, enables the achievement for the higher capacity such as 340mAh/g or more of capacity, although the cycle life therefor is as usual.

CLAIMS

1. A hydrogen absorbing alloy having a CaCu_5 type crystal structure in its principal phase, comprising La in the range of 24 to 33% by weight in the alloy, and Mg or Ca in the range of 0.1 to 1.0% by weight in the alloy.

2. A hydrogen absorbing alloy according to claim 1, further comprising 9% by weight or less of Co in the alloy.

3. A hydrogen absorbing alloy according to claim 1, further comprising 6% by weight or less of Co in the alloy.

4. A hydrogen absorbing alloy according to claim 1, wherein the Co content is 6 to 9% by weight, and the atomic ratio B/A is 5.0 to 5.25, where A represents a rare earth element including La, and B represents a rare earth element, transition metal or Al.

5. A hydrogen absorbing alloy according to claim 1, further comprising one or more selected from the group consisting of Ti, Zr and V.

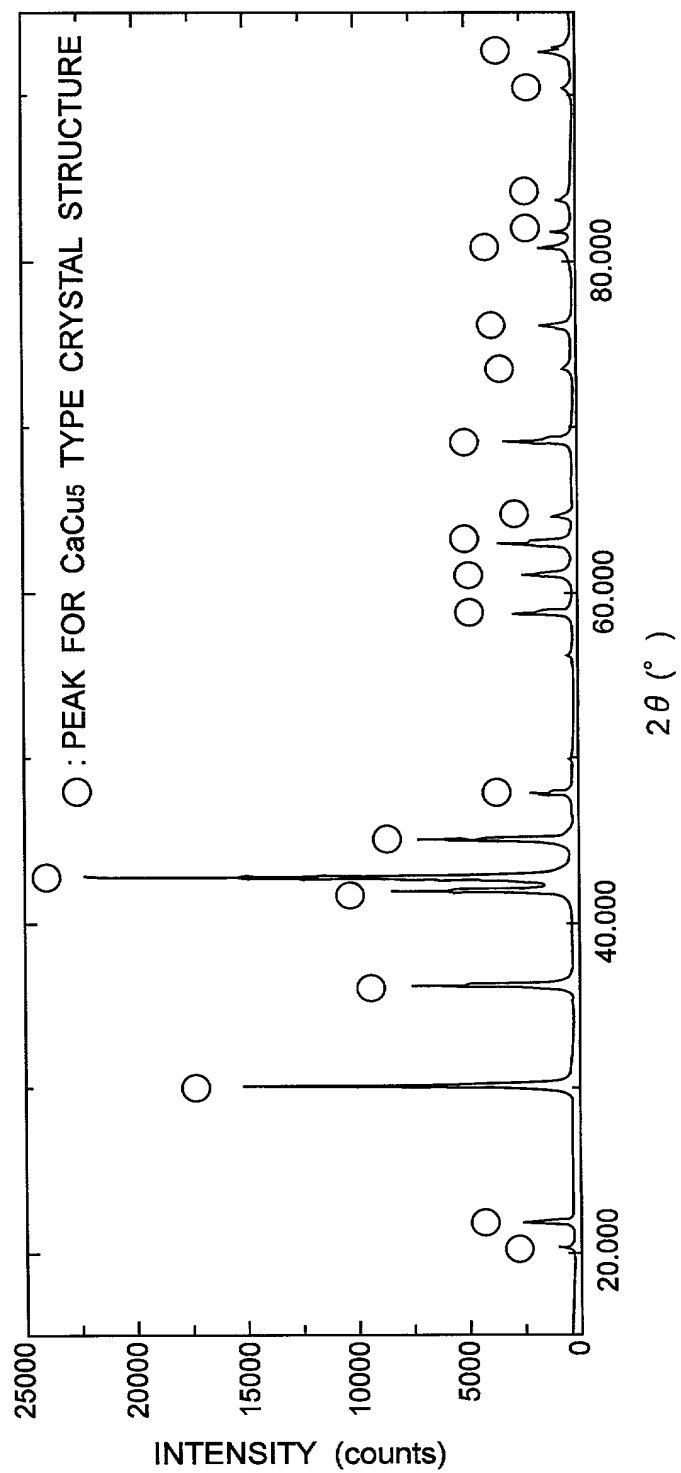
6. A hydrogen absorbing alloy having a CaCu_5 type crystal structure in its principal phase, comprising Mg and having a-axis length of 4.990 to 5.050 Å and c-axis length of 4.030 to 4.070 Å for the lattice constants in the CaCu_5 type crystal structure.

7. A hydrogen absorbing alloy according to any one of claims 1 to 4 having a-axis length of 4.990 to 5.050 Å and

ABSTRACT OF THE DISCLOSURE

An object of the present invention is to provide a hydrogen absorbing alloy which can improve a high rate discharge property while suppressing particle size reduction, exhibits cycle life characteristics equal to or higher than those of conventional alloys even when its cobalt content is decreased, and has a high capacity. Specifically, the present invention provides a hydrogen absorbing alloy having a CaCu_2 type crystal structure in its principal phase, wherein the La content in the alloy is in the range of 24 to 33% by weight and the Mg or Ca content in the alloy is in the range of 0.1 to 1.0% by weight, as well as the aforesaid alloy wherein the Co content in the alloy is not greater than 9% by weight.

FIG.1



DECLARATION AND POWER OF ATTORNEY FOR PATENT APPLICATION

Attorney Docket No. _____

As a below named inventor, I hereby declare that:

My residence, post office address and citizenship are as stated below next to my name.

I believe I am the original, first and sole inventor (if only one name is listed below) or an original, first and joint inventor (if plural names are listed below) of the subject matter which is claimed and for which a patent is sought on the invention entitled
Hydrogen Absorbing Alloy and Nickel-Metal Hydride
Rechargeable Battery

the specification of which

[X] is attached hereto

OR

[] was filed on _____ as United States Application No. or PCT International Application Number _____ and was amended on _____ (if applicable).

I hereby state that I have reviewed and understand the contents of the above-identified specification, including the claims, as amended by any amendment referred to above.

I acknowledge the duty to disclose information which is material to patentability as defined in Title 37 Code of Federal Regulations, § 1.56.

I hereby claim foreign priority benefits under Title 35, United States Code, § 119(a)-(d) or § 365(b) of any foreign application(s) for patent or inventor's certificate, or § 365(a) of any PCT international application which designated at least one country other than the United States of America, listed below and have also identified below any foreign application for patent or inventor's certificate, or of any PCT International application having a filing date before that of the application on which priority is claimed.

221990/1999	JAPAN	08/05/1999	[X] Yes [] No
Number	Country	MM/DD/YYYY Filed	Priority Claimed
2000-189040	JAPAN	06/23/2000	[X] Yes [] No
Number	Country	MM/DD/YYYY Filed	Priority Claimed
			[] Yes [] No
Number	Country	MM/DD/YYYY Filed	Priority Claimed

Table 1. Demographic characteristics of the study population	
Age (years)	
18-24	100 (10.0)
25-34	150 (15.0)
35-44	200 (20.0)
45-54	250 (25.0)
55-64	300 (30.0)
65-74	350 (35.0)
75-84	400 (40.0)
85-94	450 (45.0)
95-104	500 (50.0)
Gender	
Male	500 (50.0)
Female	500 (50.0)
Ethnicity	
White	300 (30.0)
Black	200 (20.0)
Hispanic	100 (10.0)
Asian	50 (5.0)
Other	50 (5.0)
Education	
High school or less	150 (15.0)
Some college	200 (20.0)
Bachelor's degree	250 (25.0)
Master's degree	300 (30.0)
Doctorate	350 (35.0)
Income	
<\$10,000	100 (10.0)
\$10,000-\$19,999	150 (15.0)
\$20,000-\$29,999	200 (20.0)
\$30,000-\$39,999	250 (25.0)
\$40,000-\$49,999	300 (30.0)
\$50,000-\$59,999	350 (35.0)
\$60,000-\$69,999	400 (40.0)
\$70,000-\$79,999	450 (45.0)
\$80,000-\$89,999	500 (50.0)
\$90,000-\$99,999	550 (55.0)
\$100,000+	600 (60.0)
Marital status	
Married	300 (30.0)
Single	200 (20.0)
Divorced	100 (10.0)
Widowed	50 (5.0)
Other	50 (5.0)
Health status	
Excellent	100 (10.0)
Very good	150 (15.0)
Good	200 (20.0)
Fair	250 (25.0)
Poor	300 (30.0)
Very poor	350 (35.0)
Extremely poor	400 (40.0)
Unknown	450 (45.0)
Smoking status	
Never smoker	100 (10.0)
Former smoker	150 (15.0)
Current smoker	200 (20.0)
Unknown	250 (25.0)
Alcohol consumption	
Never drinker	100 (10.0)
Former drinker	150 (15.0)
Current drinker	200 (20.0)
Unknown	250 (25.0)
Exercise frequency	
Never	100 (10.0)
Once a week	150 (15.0)
Twice a week	200 (20.0)
Three times a week	250 (25.0)
Four times a week	300 (30.0)
Five times a week	350 (35.0)
Six times a week	400 (40.0)
Seven times a week	450 (45.0)
Eight times a week	500 (50.0)
Nine times a week	550 (55.0)
Ten times a week	600 (60.0)
Stress level	
Low	100 (10.0)
Medium	150 (15.0)
High	200 (20.0)
Very high	250 (25.0)
Extremely high	300 (30.0)
Unknown	350 (35.0)

Application Number(s)	Filing Date (MM/DD/YYYY)
Application Number(s)	Filing Date (MM/DD/YYYY)

Appln. Serial No.	Filing Date	Status Patented/Pending/Abandoned
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Appln. Serial No.	Filing Date	Status Patented/Pending/Abandoned

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ENGLISH LANGUAGE DECLARATION CONTINUED

POWER OF ATTORNEY: As a named inventor, I hereby appoint the practitioners associated with the Customer Number provided below to prosecute this application and to transact all business in the Patent and Trademark Office connected therewith, and direct that all correspondence be addressed to that Customer Number:

Customer Number 826

Direct correspondence to the
attention of and telephone calls to:

Tel (704) 331-6000

Fax (704) 334-2014

Full name of (first/sole) inventor: Takao MAEDA

Inventor's

Signature: Takao Maeda Date: July 12, 2000

Residence: Takefu, Fukui-ken, Japan

Citizenship: Japan

Post Office Address: c/o Magnetic Materials Research & Development Center,
SHIN-ETSU CHEMICAL CO., LTD.,
1-5, Kitago 2-chome, Takefu, Fukui-ken, Japan

Full name of second inventor: Satoshi SHIMA

Inventor's

Signature: Satoshi Shima Date: July 12, 2000

Residence: Takefu, Fukui-ken, Japan

Citizenship: Japan

Post Office Address: c/o Magnetic Materials Research & Development Center,
SHIN-ETSU CHEMICAL CO., LTD.,
1-5, Kitago 2-chome, Takefu, Fukui-ken, Japan

Full name of third inventor: Naofumi SHINYA

Inventor's

Signature: Naofumi Shinya Date: July 12, 2000

Residence: Takefu, Fukui-ken, Japan

Citizenship: Japan

Post Office Address: c/o Magnetic Materials Research & Development Center,
SHIN-ETSU CHEMICAL CO., LTD.,
1-5, Kitago 2-chome, Takefu, Fukui-ken, Japan